

Extended Abstract of PSA-19

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## Chemical state analysis for buried Al<sub>2</sub>O<sub>3</sub> layer of Y<sub>2</sub>O<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> multi-layer film using laboratory HAXPES

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We examined the practicability and chemical state analysis of laboratory HAXPES using monochromatic Cr K $\alpha$  excitation. The acquisition time of nine min and the FWHM of 1.03 eV was obtained by Au 4f measurements. We also clarified the relationship between the oxygen gas flow rate and the chemical state of the Al<sub>2</sub>O<sub>3</sub> layer around the Y<sub>2</sub>O<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> interface for the buried Al<sub>2</sub>O<sub>3</sub> layer of Y<sub>2</sub>O<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> multi-layer film. Our findings demonstrate that laboratory HAXPES with Cr K $\alpha$  is capable of nondestructive chemical state analysis with practical throughput.

### 1. Introduction

Hard x-ray photoelectron spectroscopy (HAXPES) offers larger information depth compared to conventional x-ray photoelectron spectroscopy (XPS) using monochromatic Al K $\alpha$  ( $h\nu = 1486.6$  eV), and can perform bulk sensitive measurement [1]. HAXPES has been realized in synchrotron radiation facilities, as the photoionization cross section becomes extremely small [2], and it is not easy to secure machine time. This makes it difficult to use it for quick feedback on the manufacturing processes and quality control. Recently, laboratory HAXPES using monochromatic Cr K $\alpha$  ( $h\nu = 5414.9$  eV) has been put into practical use, and it has now become easier to measure HAXPES spectra in the laboratory [3–5].

The purpose of our study is to develop analytical techniques with practical throughput using laboratory HAXPES. In this paper, we focused on Y<sub>2</sub>O<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> multi-layer film, which is often used in electronic and optical devices. Chemical state analysis of the Al<sub>2</sub>O<sub>3</sub> layer around the Y<sub>2</sub>O<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> interface in the Y<sub>2</sub>O<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> multi-layer film sample was performed.

### 2. Experimental

We used a Y<sub>2</sub>O<sub>3</sub>(15 nm)/Al<sub>2</sub>O<sub>3</sub>(180 nm) multi-layer film sample deposited on a metal alloy substrate by magnetron sputtering in this study. Samples A, B, and C with difference oxidation states of the Al<sub>2</sub>O<sub>3</sub> layer were prepared by changing the oxygen gas flow rate at

the time of sample preparation. The oxygen gas flow rate was set to decrease in the order of sample A, B, and C.

All conventional XPS and HAXPES measurements were performed with the laboratory HAXPES system (PHI Quantes, ULVAC-PHI, Inc.) installed at the Yokohama Works of Furukawa Electric Co., Ltd.. It is equipped with a dual scanning x-ray source composed of a conventional soft x-ray source (Al K $\alpha$ ) and a hard x-ray source (Cr K $\alpha$ ). The spot size of both x-ray beams was 100  $\mu\text{m}$  (25 W at Al K $\alpha$ , 50 W at Cr K $\alpha$ ). All narrow scan spectra were taken at the pass energy of 55.0 eV, at the acceptance angle of  $\pm 20^\circ$  and the take-off angle (TOA) (from average surface of the sample) of  $45^\circ$ – $90^\circ$ . Charge neutralization was not used in any of the measurements.

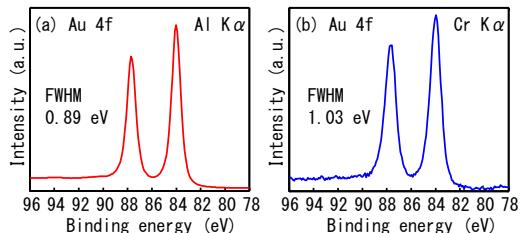
The Au plate was used to evaluate the basic performance of the laboratory HAXPES system.

### 3. Results and discussion

#### 3.1. Practicability of laboratory HAXPES

The acquisition time and the energy resolution were evaluated using the Au 4f photoelectron peak with the TOA of  $45^\circ$ . **Figure 1(a)** shows the Au 4f spectrum obtained from the Au plate by Al K $\alpha$  excitation. The acquisition time of the Au 4f spectrum was two min (0.6 s per point, 200 total points). The full width at half maximum (FWHM) of the Au 4f<sub>7/2</sub> peak was 0.89 eV.

The acquisition time of the Au 4f spectrum obtained



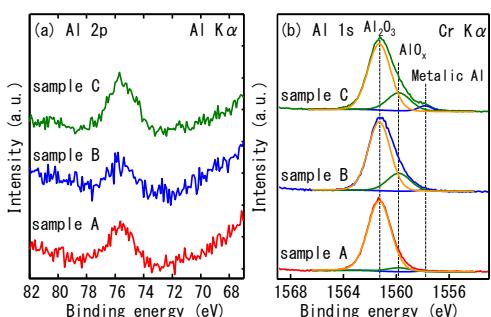
**Fig. 1** Au 4f spectra obtained from Au plate by (a) Al K $\alpha$  excitation and (b) Cr K $\alpha$  excitation.

with Cr K $\alpha$  excitation was nine min (2.7 s per point, 200 total points), as shown in **Fig. 1(b)**. The FWHM of the Au 4f $_{7/2}$  peak was 1.03 eV. The energy resolution of Cr K $\alpha$  excitation was comparable to that of Al K $\alpha$  excitation, and the acquisition time of the Cr K $\alpha$ -excited peak required for a sufficient S/N ratio did not exceed ten times that of the Al K $\alpha$ -excited peak.

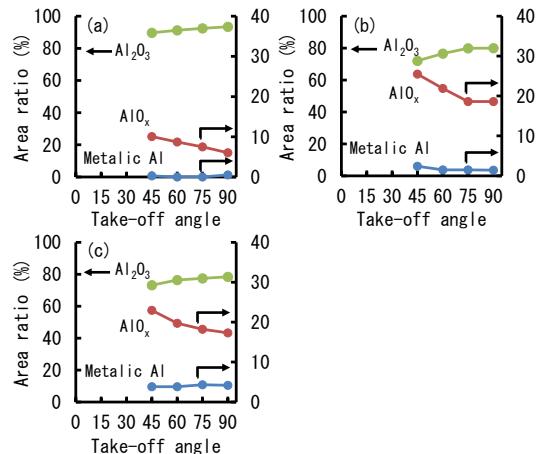
These results demonstrate that laboratory HAXPES measurement using the Cr K $\alpha$  sources can be performed with practical throughput.

### 3.2. Chemical state analysis of $\text{Al}_2\text{O}_3$ layer around $\text{Y}_2\text{O}_3/\text{Al}_2\text{O}_3$ interface

Chemical state analysis of the  $\text{Al}_2\text{O}_3$  layer in three  $\text{Y}_2\text{O}_3/\text{Al}_2\text{O}_3$  multi-layer film samples was performed in order to provide feedback to the manufacturing processes. **Figure 2(a)** shows the Al 2p spectra obtained at the TOA of 90° with Al K $\alpha$  excitation. Chemical state analysis of the  $\text{Al}_2\text{O}_3$  layer by Al K $\alpha$  excitation was difficult due to the thick  $\text{Y}_2\text{O}_3$  layer. In contrast, from Cr K $\alpha$  excitation, a high-intensity Al 1s peak was detected in three samples at the TOA of 90°, as seen in Fig. 2 (b). Al 1s spectra exhibited a peak position very close to that of the  $\text{Al}_2\text{O}_3$  phase, and had shoulders on the lower binding energy side corresponding to the sub-oxide ( $\text{AlO}_x$ ) phase. In addition, weak peaks in the metallic Al phase were also



**Fig. 2** (a) Al K $\alpha$ -excited Al 2p spectra and (b) Cr K $\alpha$ -excited Al 1s spectra obtained from sample A, B and C.



**Fig. 3** Area ratios of  $\text{Al}_2\text{O}_3$ ,  $\text{AlO}_x$  and metallic Al peaks of  $\text{Al}_2\text{O}_3$  layer in  $\text{Y}_2\text{O}_3/\text{Al}_2\text{O}_3$  multi-layer film as a function of TOA: (a) sample A, (b) sample B, and (c) sample C.

detected in samples B and C. The above three peaks were assigned in accordance with reference peak positions determined by measuring the surfaces of bare  $\text{Al}_2\text{O}_3$  and pure Al reference single film samples.

Angle-resolved measurements were performed to evaluate the Al-bonding components at the  $\text{Y}_2\text{O}_3/\text{Al}_2\text{O}_3$  interface. The area ratios of  $\text{Al}_2\text{O}_3$ ,  $\text{AlO}_x$  and metallic Al peaks in the Al 1s spectra for samples A, B, and C are plotted in **Fig. 3** as a function of the TOA. For all samples, the  $\text{AlO}_x$  component increased as the TOA was decreased, indicating that the  $\text{AlO}_x$  component exists around the  $\text{Y}_2\text{O}_3/\text{Al}_2\text{O}_3$  interface.

### 4. Conclusion

The relationship between the oxygen gas flow rate and the chemical state of the  $\text{Al}_2\text{O}_3$  layer around the  $\text{Y}_2\text{O}_3/\text{Al}_2\text{O}_3$  interface for the buried  $\text{Al}_2\text{O}_3$  layer of  $\text{Y}_2\text{O}_3/\text{Al}_2\text{O}_3$  multi-layer film was clarified by laboratory HAXPES. We have demonstrated that laboratory HAXPES with Cr K $\alpha$  is capable of nondestructive chemical state analysis with practical throughput.

### 5. References

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